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Synthesis, catalytic Si–Si dehydrocoupling, and thermolysis of polyvinylsilanes [CH₂CH(SiH₂X)]_n (X=H, Ph)

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Abstract

Clear honey-like oil $[CH_2CH(SiH_3)]_n$ (PVS) with 40% ceramic yield was crosslinked by Si–Si dehydrocoupling with the catalysts developed recently in our laboratory such as Group 4 metallocene Cp₂MCl₂/Hydride (M = Ti, Zr, Hf; Hydride = Red-Al, *N*-Selectride, Super-Hydride) combination, group 6 metal hexacarbonyl M(CO)₆ (M = Cr, Mo, W), and group 8 metallocene Cp₂M (M = Co, Ni) to give insoluble network of polyethylene–polysilane hybrids. Their themolysis in argon significantly improved ceramic yields of 62–80%. The thermogravimetric analysis ceramic yields of the crosslinked PVS with the Group 4 metallocene were higher than those with the Group 6 and 8 metal complexes. The most effective dehydrocoupling catalyst was Cp₂MCl₂/N-Selectride combination. Similarly, white sticky solid [CH₂CH(SiH₂Ph)]_n (PPVS) with 35% ceramic yield was modified by Si–Si dehydrocoupling with the Group 4 metallocene Cp₂MCl₂/Red-Al (M = Ti, Zr) combination catalysts. Thermolysis of the crosslinked PPVS under argon to 800 °C showed improved ceramic yields (47–50%), which were lower than those of crosslinked PVSs due to low degree of crosslinking and high C/Si ratio (C/Si = 8).

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1. Introduction

High performance ceramic materials have been given a great deal of interest for potential application as thermomechanical materials due to their stiffness and especially their excellent structural stability under harsh environmental extremes. Oxide ceramics exhibit creep affinity at high temperatures, limiting their application in high energy facilities and the aerospace industry [1]. By comparison, non-oxide ceramics possess increased creep stability at high temperatures, but are not thermomechanically stable in air atmosphere [2]. Silicon-containing non-oxide ceramics such as silicon carbide (SiC), silicon nitride (Si₃N₄), and silicon carbonitride (Si_xC_yN_z) have received a great deal of attention for diverse applications [3]. *Inter alia*, silicon carbide is of great interest because of its high strength, hardness, chemical resistance, semiconductivity (band gap 2.2 eV (β -SiC), 2.9 eV (α -SiC)), high-temperature stability and X-ray resistance. Thus, SiC ceramic has been widely used as reinforcing materials in advanced ceramics and composites [4]. Silicon carbide is obtained in industrial scale by Acheson process, where high-pure silicon dioxide (99.5%) and petroleum cokes are pyrolyzed together at temperatures as high as 2200–2500 °C [5]. However, the resulting SiC powder is infusible, intractable, and especially not applicable for the preparation of fibers or films.

In the mid 1970s, Yajima and coworkers [6] first showed the industrial preparation of SiC fibers, including preparation of polycarbosilane (PCS) precursors via

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the thermal rearrangement of poly(dimethylsilane) (PDMS). The PCSs obtained are melt-spun into fiber form (manufactured by Nippon Carbon Co., Tokyo, Japan and sold worldwide by Dow Corning under the trade name 'Nicalon') and are then crosslinked under air atmosphere (in the 300-350 °C temperature range) to improve the ceramic residue yield, finally pyrolyzed at 1200-1300 °C under argon atmosphere to produce β -SiC fiber along with some unwanted silicon oxycarbide and free carbon.

The advantage of polymeric precursor routes to silicon carbide is that shapes or patterns (e.g. fibers, films) may be produced at the preceramic polymer stage when the form of material is relatively easy to fabricate and may be subsequently converted by pyrolysis to an extremely firm ceramic. Great efforts have been made to tune this method with some success in the development of improved preceramic polymer for high ceramic yield and also for high phase purity. The types of preceramic polymers that have been prepared and investigated include polysilanes, PCSs, or copolymers of the two, polysiloxanes and polysilazanes, etc. [7]. The silicon atoms of these polymers for the most part are in the main chain. Notably absent among the organosilicon polymers that have been prepared for these purposes are those in which the silicon atoms are attached in pendant groups on a carbon main chain. A polymer of this type is poly(vinylsilane) (PVS), which has -SiH₃ substituent on every other carbon atom of a polyethylene backbone. PVS has been prepared by Japanese workers at Mitsui Chemical Inc., synthesized by the polymerization reactions of vinylsilane (CH2=CHSiH3) using a radical initiator (AIBN, BPO, DTBP), an anionic catalyst (ⁿBuLi, LiAlH₄, ^tBuOLi, Na/naphthalene), and a coordinated anionic catalyst (TiCl₃, TiCl₃ supported on MgCl₂) [8]. These synthetic methods are very hazardous routes to PVS because vinylsilane is a very explosive gas. Seyferth et al. reported the polymerization of vinyltrialkoxysilane and vinyltrichlorosilane initiated by ⁶⁰Co γ-radiation, followed by LiAlH₄ reduction to PVSs, and then PVSs were crosslinked by dehydrocoupling using various Group 4 metallocene compounds. Thermolysis of the crosslinked polymer in inert gas gave improved yield (70-80%) of ceramic char composed of silicon carbide together with 10-13 wt.% of free carbon [9].

Here we report the crosslinking of PVS by Si–Si dehydrocoupling with the catalysts developed recently in our laboratory such as Group 4, 6, and 8 transition metal complexes to give polyethylene–polysilane hybrids of network polymers and themolysis of the cross-linked PVSs to SiC ceramic. We also investigated the crosslinking of poly(phenylvinylsilane) (PPVS) by the Group 4 metallocene complexes and the thermolysis of the crosslinked PPVSs.

2. Results and discussion

2.1. Preparation and characterization of $[CH_2CH(SiH_2X)]_n$ (X = H, Ph)

The transparent viscous oil poly(vinylsilane) (PVS), $[CH_2CH(SiH_3)]_n$, was prepared according to the literature procedure [12].

Similarly, white semisolid poly(phenylvinylsilane) (PPVS), $[CH_2CH(SiH_2Ph)]_n$, was synthesized in a twostep reaction as summarized in Scheme 1. Radical initiator DTBP was added to a diethoxyphenylvinylsilane and kept at 150 °C while stirring for 1 day, giving poly(diethoxyphenylvinylsilane) in 95% isolated yield. Reduction of poly(diethoxyphenylvinylsilane) in diethyl ether with LiAlH₄ gave PPVS in 61% isolated yield.

The PPVS polymer was characterized by IR, ¹H-NMR, thermogravimetric analysis (TGA), and GPC. The molecular weight of PPVS measured by GPC (in THF vs polystyrene standard) was 1610 g mol^{-1} with a narrow molecular weight distribution $(M_w/M_n = 1.42)$. The IR spectrum (Fig. 1) of PPVS showed bands at 3068 (w, v_{CH}, vinyl), 2929 (w, v_{CH}, aliphatic), 2131 (s, v_{SiH}), 1590 (w, $v_{C=C}$, Ph), 1428 (s, δ_{CH} , aliphatic), 1372 (w, $\delta_{\rm CH}$, aliphatic), and 933 (s, $\delta_{\rm SiH}$). The ¹H-NMR spectrum (Fig. 1) of PPVS showed broad three peaks at δ 0.5–1.0, δ 1.0–1.3, δ 1.3–1.8 ppm assigned to head to head units of (PhSiH₂)CHCH(SiH₂Ph) and head-totail units of (PhSiH₂)CHCH, and (PhSiH₂)CHCH₂, respectively. There were resonance peaks at δ 4.0–4.4 and δ 6.8–7.8 assigned to PhSiH₂ and PhSiH₂, respectively, on top of a very weak peak at δ 5.0–6.0 ppm assigned to vinyl end group. The IR and ¹H-NMR spectra show the existence of vinyl end group. IR and ¹H-NMR spectra of PPVS were shown in Fig. 1.

2.2. Crosslinking reactions and themolysis of $[CH_2CH(SiH_2X)]_n$ (X = H, Ph)

Thermolysis properties of $[CH_2CH(SiH_2X)]_n$ (X = H, Ph) were determined by TGA. Ceramic residue yield determined by TGA of PVS was ca. 40% (black char). One of the important requirements for an idealized preceramic polymer is the presence of latent reactivity (functional substituents) to obtain thermosetting or curing properties in order to increase ceramic residue yield. Silicon-containing polymers that contain Si–H





Fig. 1. IR and ¹H-NMR spectra of PPVS.

bonds whose themolysis, due to the nature of the polymer, give only low ceramic yields, can be converted to useful preceramic precursors by appropriate chemical crosslinking treatment of their Si-H bonds because the Si-H bond is a very reactive functional group under thermal conditions or under radiation or under catalysis. For example, PCS for the Nicalon SiC-containing ceramic fiber, which contains Si-H bonds on the main backbone chain as a principal component, may be converted to a more highly cross-linked polymer by exposure to air atmosphere at 300-350 °C. Because polyvinylsilane (PVS) has reactive silyl substituents (SiH₃), it also could be crosslinked effectively by various methods. We crosslinked PVS by Si-Si dehyrocoupling reaction of the silvl groups with the catalyst developed in our laboratory such as Group 4 metallocene $Cp_2MCl_2/Hydride$ (M = Ti, Zr, Hf; Hydride = Red-Al, N-Selectride, Super-Hydride) combination, with the Group 6 metal carbonyl $M(CO)_6$ (M = Cr, Mo, W), and with the Group 8 metallocene Cp_2M (M = Co, Ni) catalysts to produce polyethylene-polysilane hybrids with extended network structure (Scheme 2).



Scheme 2.

It is assumed as shown in Scheme 2 that the insoluble cross-linked PVS could have an extended network structure composed of partially cross-linked silyl groups and fully cross-linked silyl groups in the polymer chain, judged by IR (decrease of intensity of v_{Si-H}) and TGA (increase of TGA ceramic yield) analytical data, by solubility change, and by related literatures [9–12].

The catalysts used here as a crosslinking agent of PVS, functioned effectively as dehydrocoupling catalysts of various hydrosilanes [10]. Dehydrocoupling reaction of hydrosilanes by Group 4 metallocene catalysts was first reported by Aitken et al. in 1985 [11a-c]. The dehydrocoupling reaction was further developed by other researchers [11d-e]. The dehydrocoupling reactivity of hydrosilane is well known to decrease dramatically in the order of primary > secondary > > tertiary. Therefore, it was expected that dehydrocoupling cross-linking reaction of PVS having primary silyl group (SiH₃) with various combination catalysts will take place fast at ambient temperature. The dehydrocoupling reaction rate of PVS with secondary silyl group should be much slower than that of PVS.

With the Cp₂MCl₂/Red-Al combination catalysts (M = Ti, Zr, Hf), the crosslinked polymers by dehydrocoupling of PVS were isolated in 80-90% yields which were insoluble in most organic solvents. The reactivity of dehydrocoupling on PVS decreased in the order of Ti > Zr > Hf. The TGA ceramic yield of the crosslinked PVS was obtained in the 70-72% range. The IR spectra of the crosslinked PVS showed an intense v_{Si-H} band at ca. 2150 cm^{-1} of which intensity was somewhat decreased as compared with the uncrosslinked PVS, indicating the occurrence of dehydrocoupling of silyl group. In the Cp₂MCl₂/N-Selectride and Cp₂MCl₂/ Super-Hydride combination catalysts (M = Ti, Zr, Hf), the crosslinked polymers were isolated in 80-90% yield. TGA ceramic residue yields of the crosslinked polymers were obtained in the 66-80% range for all catalysts. The reactivity of dehydrocoupling reaction on PVS decreased in the order of Ti > Zr > Hf. The effect of hydrides in the same metallocene compound were investigated. The reactivity decreased with order of Red-Al > N-Selectride > Super-Hydride for Cp_2ZrCl_2 . Dehydrocoupling crosslinking of PVS with 3 mol% of the Group 6 metal hexacarbonyl $M(CO)_6$ (M = Cr, Mo, W) catalysts was carried out at 80 °C due to lower dehydrocoupling reactivity than the Group 4 metallocene catalysts at room temperature. The crosslinked polymers were isolated in 91-97% yield. TGA ceramic yields of the crosslinked PVS were obtained in the 62-66% range. The IR spectra of the crosslinked PVS showed an intense v_{Si-H} band at ca. 2150 cm⁻¹ of which intensity was somewhat decreased as compared with the uncrosslinked PVS, implying the dehydrocoupling of silyl group. For the Group 8 metallocene Cp_2M (M = Co, Ni), the crosslinked PVSs were isolated in 90-97% yields. TGA ceramic yield of the crosslinked PVS was 64%. The reactivity of dehydrocoupling reaction on PVS decreased in the order of Ni > Co. Scanning electron microscopy (SEM) analysis of crosslinked PVS with Cp₂Ni showed gathered structure of leaflets (Fig. 2).

No evidence for an ordered structure was found in the SEM analysis of crosslinked PVS. No appreciable crystalline phase was found in XRD analysis. This polymer has very low surface area (BET method, N_2) of 0.24 m² g⁻¹ due to low degree of crosslinking. The characterization results of PVS are summarized in Table 1.

As shown in Table 1, TGA ceramic yields of the crosslinked PVS with the Group 4 metallocene were



Fig. 2. SEM image of crosslinked PVS (crosslinked with $\mbox{Cp}_2\mbox{Ni})$ at magnification of 4000.

relatively higher than those with the Group 6 and 8 metal complexes due to higher dehydrocoupling reactivity. Among the Group 4 metallocene combination catalysts, the Cp₂MCl₂/N-Selectride combination provided highest TGA ceramic yield. Degree of crosslinking is one of the most important factors that affect ceramic yield of preceramic polymers. Therefore, based on TGA ceramic yield, the most effective crosslinking agent is Cp₂MCl₂/N-Selectride combination catalysts (M = Ti, Zr, Hf). TGA thermograms of PVS and crosslinked PVS were shown in Fig. 3.

PPVS containing secondary silvl phenyl pendant group could be also crosslinked by various dehydrocoupling catalysts. We examined the crosslinking reaction of PPVS with dehydrocoupling $Cp_2MCl_2/Red-Al$ (M = Ti, Zr) combination catalysts. As expected, dehydrocoupling crosslinking of PPVS with Cp2MCl2/Red-Al combination catalysts took place very slowly at ambient temperature and upon heating at 80 °C the reaction mixture became slowly gelatinous. TGA ceramic yields of the crosslinked polymers were obtained in 48% for Ti and 51% for Zr. The relatively low TGA ceramic yield of the crosslinked PPVS is due to low degree of crosslinking and high C/Si ratio (C/Si = 8). The reactivity of dehydrocoupling reaction on PPVS decreased with the order of Ti>Zr. TGA thermograms of PPVS and crosslinked PPVS were shown in Fig. 4.

In summary, PVSs $[CH_2CH(SiH_2X)]_n$ (X = H: PVS; X = Ph: PPVS) were prepared by polymerization of $CH_2 = CHSi(OEt)_2X'$ (X' = OEt, Ph) in the presence of di-tert-butyl peroxide radical initiator followed by reaction of $[CH_2CHSi(OEt)_2X]_n$ (X' = OEt, Ph) with lithium aluminum hydride in diethyl ether. PVS and PPVS were modified by Si-Si dehydrocoupling crosslinking with the Group 4 metallocene Cp₂MCl₂/hydride (M = Ti, Zr, Hf; Hydride = Red-Al, N-Selectride,Super-Hydride), Group 6 metal hexacarbonyl M(CO)₆ (M = Cr, Mo, W), and Group 8 metallocene Cp_2M (M = Co, Ni) catalysts to give polyethylene-polysilane hybrids with extended network structure. Thermolysis of the crosslinked polymers under nitrogen to 800 °C showed improved TGA ceramic residue yields. In the dehydrocoupling modification, the Group 4 metallocenes were more effective catalyst than the Group 6 and 8 metal complexes. The most effective dehydrocoupling catalyst was Cp₂MCl₂/N-Selectride combination.

3. Experimental

All reactions and manipulations were performed under prepurified dry nitrogen atmosphere using Schlenk technique. Solvents such as diethyl ether and toluene were used after distilling from Na/benzophenone. Infrared spectra were obtained using a Nicolet 540 FT-IR spectrometer. ¹H-NMR spectra were acquired

Table 1 Characterization results of dehydrocoupling crosslinking of PVS

Dehydrocoupling catalysts	М	Yield (%) ^a	Gelation time	Ceramic yield (%) at 800 $^{\circ}\mathrm{C}^{\mathrm{\ b}}$
Uncrosslinked PVS	_	_	_	40
Group 4 °				
Cp ₂ MCl ₂ /Red-Al	Ti	90	1 min	71
	Zr	80	5 min	71
	Hf	83	25 min	72
Cp2MCl2/N-Selectride	Ti	91	4.5 h	79
	Zr	80	5.0 h	80
	Hf	88	5.2 h	78
Cp ₂ MCl ₂ /Super-Hydride	Ti	90	5.5 h	70
	Zr	87	12.5 h	67
	Hf	83	24 h	66
Group 6 ^d				
M(CO) ₆	Cr	97	20 h	63
	Мо	94	23 h	65
	W	91	7.5 h	62
Group 8 °				
Cp_2M	Co	97	9 h	64
	Ni	90	45 min	64

^a Insoluble part in common organic solvents.

^b Measured with TGA up to 800 °C.

^c [catalyst]/[PVS] = 0.02.

^d [catalyst]/[PVS] = 0.03.

using a Bruker ASX 32 (300 MHz) spectrometer using $CDCl_3/CHCl_3$ as a reference at 7.24 ppm downfield from TMS. TGA was performed using a Shimadzu TGA-50 Series Thermal Analysis System with a program rate of 10 °C min⁻¹ in argon atmosphere (flow rate, 20 ml min⁻¹). Ceramic residue yield was obtained as the weight percentage of the char remaining after completion of the heating cycle initiated by 100% of the original sample weight. Average molecular weights of

the polymer were determined by Waters Millipore GPC liquid chromatograph with four Ultrastyragel GPC columns series (sequence 500, 10^3 , 10^4 , 10^5 Å columns) calibrated with monodispere polystyrene standards. Scanning electron microscope (SEM) micrograph of the crosslinked polymers were obtained on a JEOL JXA 840. BET surface area was measured using Quntachrome Autosorb-1 Porosimeter. Cp₂MCl₂ (M = Ti, Zr, Hf), M(CO)₆ (M = Cr, Mo, W), Cp₂M



Fig. 3. TGA thermograms of PVS and crosslinked PVS at a heating rate of 20 °C min⁻¹ under an argon flow (20 ml min⁻¹).



Fig. 4. TGA thermograms of PPVS and crosslinked PPVS at a heating rate of 10 °C min⁻¹ under an argon flow (20 ml min⁻¹).

(M = Co, Ni), Red-Al (sodium bis(2-methoxyethoxy)aluminum hydride; 65 wt.% in toluene), Super-Hydride (lithium triethylborohydride; 1.0 M in THF), N-Selectride (sodium tri-sec-butylborohydride; 1.0 M in THF), vinyltriethoxysilane, di-tert-butyl peroxide (DTBP), LiAlH₄ and HCl (12 M) were purchased from Aldrich Chemical Co. and were used without further purification. Diethoxyphenylvinylsilane was prepared by reaction of phenylmagnesium bromide and vinyltriethoxysilane in THF. PVS [¹H-NMR (300)MHz, CDCl₃, ppm): δ 0.6-0.9 (1H, CH), 1.0-1.4 (2H, CH₂), 1.4–1.9 (2H, CH₂), 3.4–3.7 (3H, SiH₃); IR (KBr film, neat, cm⁻¹): 2149 (v_{Si-H}); GPC (in THF vs polystyrene standard): $M_{\rm w} = 1300 (M_{\rm w}/M_{\rm n} = 2.5)$; TGA ceramic residue yield: 40%) was prepared according to the literature procedure [12].

3.1. Preparation of PPVS

A 100-ml Schlenk flask was filled with vinyltriethoxysilane (20 g, 90.0 mmol) and DTBP (0.10 g, 0.5 wt.%) under nitrogen stream. Heating at 150-155 °C for 1 day resulted in a clear, colorless, viscous liquid which was then kept under reduced pressure while heating to 100 °C for 6 h. A solution of 19.0 g (85.5 mmol equivalent) of the resulting residue in 100 ml diethyl ether was added dropwise to an ice-cooled suspension of 4.87 g (128 mmol) of LiAlH₄ in 250 ml of diethyl ether. The reaction mixture was stirred and allowed to warm to room temperature. After it had been stirred for 2 days, the mixture was refluxed for 3 h. The ether solution was then slowly added to ice-cooled 200 ml 2 M-HCl aqueous solution while stirred. The slurry was extracted with diethyl ether in various portions. The ether phases were combined, rinsed with water, were dried over anhydrous MgSO₄ and subsequently removal of ether in vacuo left PPVS 7.0 g (61%) as a white sticky solid. ¹H-NMR (300 MHz, CDCl₃, ppm): δ 0.5–1.0 (1H, CH), 1.0–1.3 (1H, CH), 1.3–1.8 (2H, CH₂), 4.0– 4.4 (2H, SiH₂), 6.8–7.8 (5H, Ph); IR (KBr pellet, cm⁻¹): 3068 (w), 2969 (w), 2929 (w), 2131 (s, $v_{\text{Si}-\text{H}}$), 1590 (w), 1428 (s), 1372 (w), 1184 (w), 1119 (s), 1062 (s), 933 (s), 835 (s) 700 (s); GPC (in THF vs polystyrene standard) $M_{\text{w}} = 1611 (M_{\text{w}}/M_{\text{n}} = 1.42)$; TGA ceramic residue yield: 35%.

3.2. Crosslinking reactions of PVS catalyzed by $Cp_2MCl_2/Red-Al$ (M = Ti, Zr, Hf)

The following procedure is the representative of the crosslinking reactions with the Group 4 metallocene $Cp_2MCl_2/Red-Al$ (M = Ti, Zr, Hf) combination catalysts. To a 100-ml Schlenk flask were added Cp₂TiCl₂ (0.030 g, 0.12 mmol) and Red-Al (0.035 ml, 0.12 mmol) under nitrogen atmosphere. The mixture was stirred at room temperature for 3 min. PVS solution (0.35 g/0.3 ml in toluene) was then slowly injected to the mixture. The reaction mixture turned into deep green immediately evolving hydrogen gas and was solidified with yellowbrown coloration after 1 min. The catalyst was inactivated 1 day later by exposure to the air for a few min. The reaction product was washed well with diethyl ether in various portions and dried under reduced pressure at 50 °C for 1 day to give 0.314 g (90% yield) of pale yellow solid (TGA ceramic residue yield: 71%).

3.3. Crosslinking reactions of PVS catalyzed by $Cp_2MCl_2|N$ -Selectride (M = Ti, Zr, Hf)

The following procedure is the representative of the crosslinking reactions with the Group 4 metallocene Cp_2MCl_2/N -Selectride (M = Ti, Zr, Hf) combination catalysts. PVS (0.35 g, 0.12 mol) in 0.30 ml of toluene was added to a Schlenk flask containing Cp_2TiCl_2

(0.030, 0.12 mmol) and N-Selectride (0.024 ml, 0.24 mmol). The reaction mixture turned into deep green immediately evolving hydrogen gas and was solidified with dark green coloration after 4.5 h. After 1 day the catalyst was inactivated by exposure to the air for a few minutes. The reaction product was rinsed well with diethyl ether in various portions and dried under reduced pressure at 50 °C for 1 day to give 0.317 g (91% yield) of pale yellow solid (TGA ceramic residue yield: 79%).

3.4. Crosslinking reactions of PVS catalyzed by $Cp_2MCl_2/Super-Hydride$ (M = Ti, Zr, Hf)

In a typical reaction, to a 100-ml Schlenk flask were added Cp_2TiCl_2 (0.030 g, 0.12 mmol) and Super-Hydride (0.024 ml, 0.24 mmol) under nitrogen atmosphere. The mixture was stirred at room temperature for 3 min. PVS solution (0.35 g/0.3 ml in toluene) was then slowly injected. The reaction mixture turned into deep green immediately evolving hydrogen gas and was solidified with orange coloration after 5.5 h. The catalyst was inactivated 24 h later by exposure to the air for a few min. The reaction product was washed well with diethyl ether in various portions and dried under reduced pressure at 50 °C for 1 day to afford 0.314 g (90% yield) of pale yellow solid (TGA ceramic residue yield: 70%).

3.5. Crosslinking reactions of PVS catalyzed by $M(CO)_6$ (M = Cr, Mo, W)

As a typical reaction, to a 100-ml Schlenk flask was added $Cr(CO)_6$ (0.026 g, 0.18 mmol) under nitrogen atmosphere, and PVS (0.35 g, 6.02 mmol) in 2 ml of dioxane was then charged. The reaction mixture was slowly heated up 80 °C in order to minimize the sublimination of $Cr(CO)_6$ and was solidified with light yellow coloration after 20 h. After 24 h, the resulting mixture was rinsed well with diethyl ether in various portions and dried under reduced pressure at 50 °C for 1 day to provide 0.340 g (97% yield) of light yellow solid (TGA ceramic residue yield: 63%).

3.6. Crosslinking reactions of PVS catalyzed by Cp_2M (M = Co, Ni)

The following procedure is the representative of the crosslinking reactions with the Group 8 metallocene Cp_2M (M = Co, Ni) catalysts. To a 100-ml Schlenk flask was added Cp_2Ni (0.023 g, 0.12 mmol) under nitrogen atmosphere, and then, PVS solution (0.35 g/0.3 ml in toluene) was slowly injected. The reaction mixture turned into light green immediately evolving hydrogen gas and was solidified with brown coloration after 45 min. The reaction product was washed well with diethyl

ether in various portions and dried under reduced pressure at 50 $^{\circ}$ C for 1 day to yield 0.314 g (90% yield) of beige solid (TGA ceramic residue yield: 64%).

3.7. Crosslinking reactions of PPVS catalyzed by $Cp_2MCl_2|Red-Al \ (M = Ti, Zr)$

As a typical reaction, to a 100-ml Schlenk flask were charged Cp₂TiCl₂ (0.013 g, 0.053 mmol) and Red-Al (0.016 ml, 0.053 mmol) under nitrogen atmosphere, and PPVS (0.35 g, 2.65 mmol) in 1 ml of toluene was slowly syringed. The reaction mixture turned into deep brownish green immediately evolving hydrogen gas. After 5 min, the reaction mixture was slowly heated up to 80 °C and was solidified with brownish yellow coloration after 30 h. The resulting mixture was rinsed well with diethyl ether in various portions and dried under reduced pressure at 50 °C for 1 day to give 0.330 g (94% yield) of pale yellow solid (TGA ceramic residue yield: 48%).

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